Electron Transport in a Multichannel One-Dimensional Conductor: Molybdenum Selenide Nanowires

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We have measured electron transport in small bundles of identical conducting molybdenum selenide nanowires where the number of weakly interacting one-dimensional chains ranges from 1 to 300. The linear conductance and current in these nanowires exhibit a power-law dependence on temperature and bias voltage, respectively. The exponents governing these power laws decrease as the number of conducting channels increase. These exponents can be related to the electron-electron interaction parameter for transport in multichannel 1D systems with a few defects.

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Interacting electrons in one-dimensional (1D) metals constitute a Luttinger liquid (LL) [1], in contrast to a Fermi liquid (FL) in three-dimensional (3D) metals. Transport properties of 1D conductors are strongly modified as adding an electron to a 1D metal requires changing the many-body state of its collective excitations. This results in a vanishing electron tunneling density of states at low energy. Power-law dependent suppression in tunneling conductance has been observed in many systems, including fractional quantum Hall edge states [2], single and multiwalled carbon nanotubes [3–5], bundles of NbSe₄ nanowires [6], and conducting polymers [7]. A crossover from a truly 1D LL to a 3D FL is expected as 1D conductors are coupled together, increasing the number of weakly interacting channels [8,9]. This transition, however, has not been observed in the above (quasi-) 1D systems due to the experimental difficulty in preparing identical conducting quantum wires to form conductors with a few weakly interacting channels. In this Letter, we report temperature and bias dependent electric transport measurements on small bundles of molybdenum selenide (MoSe) nanowires [10–12], whose diameter ranges from 1–15 nm. These nanowires, which consist of bundles of weakly interacting and electrically identical 1D MoSe molecular chains, show a power-law dependent tunneling conductance. The exponent governing the power law decreases as the bundle diameter increases, indicating a transition from 1D to 3D bulk transport with increasing number of conducting channels.

Crystalline bundles of MoSe chains are obtained from the dissolution of quasi-1D Li₂Mo₆Se₆ crystals in polar solvents. Single crystal Li₂Mo₆Se₆ was prepared as described previously [13]. X-ray diffraction analysis showed hexagonally close packed molecular MoSe chains with a lattice spacing \( a₀ = 0.85 \) nm, separated by Li atoms [Fig. 1(a)]. Atomic scale bundles of MoSe nanowires were produced from \( \sim 100 \) \( \mu \)M solutions of Li₂Mo₆Se₆ in anhydrous methanol. The solutions were then spun onto degenerately doped Si/SiO₂ substrates with lithographically patterned electrodes in a nitrogen atmosphere. Typically, 35 nm thick Au electrodes with a 5 nm Cr adhesion layer separated by \( \sim 1 \) \( \mu \)m were used to contact randomly deposited nanowires. Figs. 1(b) and 1(c) show atomic force microscope (AFM) images of typical devices. The two-probe resistance of such a device, which ranged from \( \sim 100 \) kΩ to 100 MΩ at room temperature, was measured in a cryostat with a continuous flow of helium. A degenerately doped silicon substrate, underneath a \( L_{ox} = 300 \) nm thick silicon dioxide dielectric layer, served as a back gate to modulate the charge density in the nanowires. Once transport measurements were complete, the wire diameter, \( D \), was determined from an AFM height profile.

Figure 2 shows the conductance \( (G) \) normalized by its room temperature value as a function of temperature \( (T) \) for a representative subset of the samples investigated [14]. We applied a small bias voltage, \( (V \ll k_B T/e) \), to stay in the linear response regime for this measurement. Notably, the mesoscopic scale samples \( (D < 20 \text{ nm}, L \sim 1 \mu \text{m}) \) exhibit more than 2 orders of magnitude conductance

\[ \frac{G}{G_{RT}} \sim T^{-\alpha} \]

\[ \frac{G}{G_{RT}} \sim V^{-\beta} \]

FIG. 1 (color online). (a) Structural model of a 7-chain MoSe nanowire along with the triangular Mo₃Se₃ unit cell. (b) and (c) AFM height images of MoSe nanowires between two Au electrodes. The wire heights are 7.2 nm and 12.0 nm, respectively. Scale bar = 500 nm.

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decrease with decreasing temperature in the measured temperature range, unlike the samples in the bulk limit $(D > 1 \, \mu \text{m}, L \sim 100 \, \mu \text{m})$, which exhibit a bulk metallic behavior, i.e., the conductance increases with decreasing temperature. A power-law dependence, $G \sim T^n$, is evident in these mesoscopic samples, where the exponent $\alpha$ can be readily extracted from the slope of the least-squares fit line in the double logarithm plot. For most of the samples, $G(T)$ can be expressed by a single $\alpha$ within the experimentally accessible conductance range [15]. However, for some wires with a relatively high conductance (>1 $\mu$S) at room temperature (Fig. 2, ▲ and ▼), an abrupt change in the exponent at low temperatures was observed. In this low temperature regime, the conductance varied with the gate voltage (Fig. 2, inset) and the exponent depended on the applied gate voltage, $V_g$, due to Coulomb charging effects. Above this Coulomb charging temperature, a general trend of decreasing $\alpha$ with increasing $D$ is found for all mesoscopic samples studied. This trend will be discussed further later in this Letter.

We now consider several possible explanations for seeing a decreasing conductance with decreasing temperature. For example, one expects $\ln(G) \propto -1/T$ for barrier activated transport. For highly defective wires, one expects $\ln(G) \propto -1/T^\delta$ due to variable range hopping between localized states in the wire, where $\delta$ can range from 1/4 for a 3D wire to 1/2 for a 1D wire [16]. However, neither of these models fit our data as the conductance follows a power law remarkably, irrespective of wire diameter. Another possible explanation is a nonmetallic behavior associated with a Peierls transition, which opens up an energy gap at the Fermi level. However, from the conductance measurements on the bulk quasi-1D crystals $(D > 1 \, \mu \text{m})$ and also from previous scanning tunneling microscopy work on similar nanowires [12], we observe no evidence for such a gap opening at temperatures down to 5 K, consistent with recent band calculations [17]. Alternatively, a power-law dependent tunneling conductance is predicted for tunneling into a Luttinger liquid [1], into 1D Wigner crystals [18], and into a highly disordered system where the electron mean free path is comparable with the wire diameter [19]. We first rule out a 1D Wigner crystal model since the Coulomb interactions, which are screened by the back gate, are not long ranged. We also eliminate the scenario for a strongly disordered system by estimating the electron mean free path, $l_c$, in the nanowire. We can infer $l_c$ indirectly from the effective wire length, $L_{eff}$, obtained from the dependence of the Coulomb charging energy on the gate voltage. Here, the charging energy $E_c = e^2/C$, where the wire capacitance $C \approx 2\pi r e L_{eff}/(ln(4L_{ox}/D))$. For the nanowire device shown in the inset of Fig. 2, the estimated charging energy $E_c$ is 5–10 meV from the conductance data measured in $V$ and $V_g$ (not shown), from which we obtained $L_{eff} \sim 0.3–0.6 \, \mu \text{m}$. With this estimate, we rule out the possibility of having a highly disordered system. Moreover, the fact that the measured resistance is larger than ~100 k$\Omega$ and is not directly correlated with the wire diameter indicates that transport is dominated by tunneling, rather than intrinsic resistivity of the wire itself. Thus a model concerning tunneling into a relatively clean LL is a more likely description of the observed transport phenomena [20].

Further support for the LL-like transport in the MoSe nanowires can be found in the bias dependence of the conductance in the nonlinear response regime. According to the LL model in a tunneling regime [1], the bias voltage dependent transport current, $I(V)$, has a transition between an Ohmic behavior, i.e., $I \propto V$ in the low bias regime $(V \ll k_BT/e)$, and a power-law behavior with an exponent $\beta$, i.e., $I \propto V^{\beta+1}$ in the high bias regime $(V \gg k_BT/e)$. The inset of Fig. 3 shows typical $I(V)$ data measured in a mesoscopic wire with the applied bias voltage ranging over more than 3 orders of magnitude at different temperatures. A transition between Ohmic and power-law behavior is observed as $V$ increases.

Similar to the dependence of the exponent alpha to the diameter, we also found that the exponent $\beta$ depended strongly on $D$. Table I lists $\alpha$, $\beta$, and $D$ for 13 samples. In general, we find that $\alpha$ decreases monotonically as $D$ increases. Based on the relation between $\alpha$ and $\beta$, we can categorize our samples into two distinct groups: group (I) where $\alpha = 2\beta$; and group (II) where $\alpha = \beta$. In our experiments, the majority of samples (10 out of 13 in Table I) belong to group (I), while only a few samples (3 out of 13 in Table I) belong to group (II) [21].

For a clean LL without defects, these two exponents are expected to be identical, i.e., $\alpha = \beta$ [1], as observed in
observed in our group (I) samples [22]. This argument does not hold, however, if there are no defects in the wire, or in the extreme of strong defects that dominate transport within the experimentally accessible range of $T$ and $V$. For such samples, $\alpha = \beta$, as it is found in group (II) samples.

The power-law behaviors in $T$ and $V$ allow us to scale $I(V, T)$ into a single curve [3,23]. Considering the above arguments, we can modify the scaling formula of a clean LL transport model to include the two exponents $\alpha$ and $\beta$ as

$$I = I_0 T^{\alpha+1} \sinh\left(\gamma eV/(2k_B T)\right) \left[1 + (\beta/2 + \gamma eV/(2k_B T))^2\right],$$

where $I_0$ and $\gamma$ are constants independent of $T$ and $V$. Physically, $\gamma$ represent the ratio between the voltage across a dominantly resistive junction at high bias to the applied bias voltage [3]. As shown in Fig. 3, the series of $I(V)$ curves measured at different temperatures for the same sample collapse remarkably well onto a single curve described by Eq. (1) over the entire measured temperature range by plotting $I/T^{\alpha+1}$ against $\gamma eV/k_B T$ with only one fitting parameter $\gamma$. For the data in Fig. 3, $\gamma$ is $0.25 \pm 0.1$, implying that there are probably four barriers of approximately equal resistance over which the applied bias voltage is distributed.

Finally, we now discuss the dependence of the exponents on the wire diameter $(D)$ in order to elucidate the transition from a few channel 1D transport to the 3D transport limit. For this purpose, we focus on the samples with $\alpha = \alpha_{LL-LL}$ [i.e., group (I) samples and the group (II) samples with $\gamma = 1$]. In Fig. 4, we show the measured $\alpha$ plotted against $D$. Since $N \propto D^2$, where $N$ is the number of channels in the wire including the spin degree of freedom [24], the observed rapid decrease of $\alpha$ for large diameter nanowire bundles indicates a crossover from 1D behavior to 3D transport ($\alpha = 0$). Employing the electron-electron interaction parameter for a single chain $(N = 2)$, $g$, the end tunneling exponent $\alpha_{LL-LL}$ for an $N$ channel LL wire can be expressed as $[8,25]$.

$$\alpha_{LL-LL} = \frac{2}{N} [(1 + NU)^{1/2} - 1],$$

where $U \approx 2 g^2$. We fit this equation to our data using $g$ as a single fitting parameter. A good agreement with the

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**TABLE I.** Measured exponents $\alpha$ and $\beta$ determined from the temperature and bias dependent conductance measurement, along with the wire diameter $(D)$ as determined by AFM and the number of channels including spin $(N)$ calculated from $D$. Wires indicated by an asterisk (*) have $\alpha \approx \beta$ but for all other wires, $\alpha \approx 2\beta$.

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<tbody>
<tr>
<td>$\alpha$</td>
<td>6.6</td>
<td>5.2</td>
<td>4.3</td>
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<tr>
<td>$\beta$</td>
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<td>4.9</td>
<td>2.1</td>
<td>1.9</td>
<td>1.0</td>
<td>0.72</td>
<td>1.2</td>
<td>1.0</td>
<td>0.8</td>
<td>1.09</td>
<td>0.6</td>
<td>0.90</td>
<td>0.32</td>
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<tr>
<td>$D$</td>
<td>$0.8 \pm 0.5$</td>
<td>$2.1 \pm 0.3$</td>
<td>$3.0 \pm 0.3$</td>
<td>$3.5 \pm 0.2$</td>
<td>$5.0 \pm 0.5$</td>
<td>$5.3 \pm 1.0$</td>
<td>$6.1 \pm 0.5$</td>
<td>$7.2 \pm 0.5$</td>
<td>$7.3 \pm 1.4$</td>
<td>$7.4 \pm 1.5$</td>
<td>$10.3 \pm 0.4$</td>
<td>$12.0 \pm 0.7$</td>
<td>$15.7 \pm 1.1$</td>
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<tr>
<td>$N$</td>
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<td>22</td>
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<td>62</td>
<td>70</td>
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<td>134</td>
<td>138</td>
<td>268</td>
<td>362</td>
<td>620</td>
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The solid line is a fit to Eq. (2).

Experimental observation was obtained for \( g = 0.15 \) (solid line in Fig. 4). For a screened Coulomb interaction, \( g \) can be estimated by \( g \approx \sqrt{1/(e^2 \ln(4L_{ox}/a_0)/\pi \hbar v_F \kappa)} \) [8], where \( v_F \) is the Fermi velocity of a single chain and \( \kappa \) is the dielectric constant of silicon dioxide. From the value of \( g \), we deduce that \( v_F \approx 3 \times 10^4 \) m/s. We note here that this value is smaller than the value obtained from recent band calculation \( (4 \times 10^5 \) m/s) [17], indicating that a static screening picture considered in this model might be too simplistic. Further theoretical considerations including the effect of impurities and interchain hopping are needed to elucidate strongly interacting electrons in these 1D channels.

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14. The mesoscopic wires were susceptible to structural deformations locally at the wire/electrode junctions. Because of the invasiveness of the electrodes, multiterminal measurements were not possible.
15. Since \( G \) decreases rapidly as \( T \) decreases in the nanowire samples, most of our \( G(T) \) data were limited at low \( T \) by our current sensitivity \( (10^{-11} \) A).
20. We note that the environmental Coulomb blockade (ECB) model (see, for example, G.-L. Ingold and Y. V. Nazarov, cond-mat/0508728) might be applied to our experiment. However, its validity is limited to the large \( N \) limit [8], where the LL and ECB models do not offer any observable differences in our experimental setup; we thus consider only the LL model in this paper.
21. Group (II) samples can be subdivided further into a group with notable defects such as the one shown in Fig. 1(b) \( (\gamma \approx 1) \) and a group without strong defects \( (\gamma \approx 1/2) \), where the parameter \( \gamma \) is defined in Eq. (1).
22. For temperatures below 300 K, the deviation of the measured \( \alpha \) from \( \alpha_{\text{FL-LL}} \) is less than 10% as long as \( R_{\text{Lead-Wire}}/R_{\text{Defect}} \) is between \( \sim 0.1 \)–10 where \( R_{\text{Lead-Wire}} \) is the total resistance between the wire and the Au lead and \( R_{\text{Defect}} \) is the total resistance of all defects. A similar tolerance range for \( R_{\text{Lead-Wire}}/R_{\text{Defect}} \) is also found for \( \beta \approx \alpha_{\text{FL-LL}} \). We have observed a few devices outside of this tolerance limit, which shows an intermediate bias ranges where LL – LL tunneling is dominant at low temperature.
23. L. Balents, cond-mat/9906032.
24. We use \( N = 2fD^2/a_0^2 \) where \( f \), the filling factor, is 0.91, assuming a hexagonal close packing of the chains.